Preparation and electrochemical behavior of methylene blue intercalated into layered niobate $K_4Nb_6O_{17}$

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Abstract $K_4Nb_6O_{17}$ nano-layered compound was obtained by solid-phase synthesis and then methylene blue (MB) was intercalated into layered niobate $K_4Nb_6O_{17}$ interlayer I by a two-step guest-guest exchange method using the intercalation compound, methyl viologen (MV^{2+})– $K_4Nb_6O_{17}$, as precursor. The optically transparent MB^+ – $K_4Nb_6O_{17}$ nanocomposite thin film has been characterized by XRD, IR, TGA, elemental analysis, UV, and electrochemical measurements. It was estimated that the intercalated MB^+ ions are mainly aggregated. The cyclic voltammogram of the MB^+ – $K_4Nb_6O_{17}$ nanocomposite thin film exhibited a fine diffusion-controlled cathodic process, which hints the possibility of being utilized as an electrode modifying material.

Introduction

Nanostructured layered metal oxides are a kind of amazing materials with unique two-dimensional structure. Structural, textural, and compositional modifications of layered metal oxides have been investigated to develop new materials with tailored properties, such as catalysis [1], photocatalysis [2–4], electrooxidation [5] and photoluminescent behaviors [6, 7], etc. $K_4Nb_6O_{17} \cdot 3H_2O$ is a unique

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semiconductor material possessing two alternative interlayer spaces (interlayer I and II) formed by [Nb₆O₁₇]⁻ layers, between which hydrated and non-hydrated K⁺ ions exist to maintain the charge balance [8, 9]. The hydrated K⁺ ions in interlayer I can be directly exchanged by both monovalent and multivalent ions [10, 11], as is shown in Fig. 1. Intercalation reactions of K₄Nb₆O₁₇ with metal ions [12, 13], methylviologen [10, 14–17], $[Ru(phen)_3]^{2+}$ ions [18], $[Ru(bpy)_3]^{2+}$ [19-23], rhodamine 6G (R6G) dye [24], porphyrin [25, 26] have been achieved, the photocatalytic, photoelectrochemical, and photo-induced electron-transfer behaviors of the intercalation nanocompound have been extensively investigated. However, although the electrochemical behaviors of the intercalated ions have been studied, the prospective utilization of the nanocomposites as modified electrodes is not widely discussed.

Methylene blue (MB) is a kind of basic dye with plane structure (Fig. 2), which is widely used for electrochemical applications, such as catalyst-mediator in electrochemical biosensors. To overcome the water-soluble disadvantages, MB has been immobilized in various matrices such as silicate [27], barium phosphate [28], TiO₂ [29], mordenite-type zeolite [30], zirconium phosphate [31], layered manganese oxide [32]. MB⁺ intercalated K₄Nb₆O₁₇ has been synthesized by ion-exchange [33] and electrostatic self-assembly deposition (ESD) [34] method, but the electrochemical behavior of intercalated MB⁺ was not reported in detail.

In the present work, MB was intercalated into interlayer I of $K_4Nb_6O_{17}$ by host-guest ion exchange method by use of layered $MV^{2+}-K_4Nb_6O_{17}$ as precursor. The $MB^+-K_4Nb_6O_{17}$ nanocomposite was characterized by X-ray diffraction, IR, TGA, elemental analysis, and UV spectroscopy. The electrochemical behaviors of the $MB^+-K_4Nb_6O_{17}$ hybrid thin film electrode were studied.

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Fig. 1 The structural model of $K_4 N b_6 O_{17} \cdot 3 H_2 O$ showing two different interlayers

Experimental

Analytical Nb_2O_5 and KOH were purchased from Sinopharm Chemical Reagent Co., Ltd., methylviologen chloride and methylene blue were purchased from Tokyo Kasei, all reagents were used without further purification.

The layered compound, $K_4Nb_6O_{17} \cdot 3H_2O$, employed here was prepared by calcination of a 2.1:3.0 molar mixture of K_2CO_3 and Nb_2O_5 at 1100 °C for 10 h, according to the procedures reported by Nassau et al. [35]. Figure 1 shows the layered structure of $K_4Nb_6O_{17} \cdot 3H_2O$. $K_4Nb_6O_{17}$ consists of octahedral units of NbO₆, which form a twodimensional layered structure via bridging oxygen atoms. The layers are negatively charged, and K^+ ions exist between the layers to compensate for the negative charges of the layers.

The intercalation compound of $K_4Nb_6O_{17}$ with MB⁺ is generally very difficult to prepare by a direct reaction because of the bulkiness of MB⁺. A two-step intercalation was thus attempted by adopting methylviologen- $K_4Nb_6O_{17}$ intercalation compounds as the intermediate. The methylviologen cation (MV²⁺) can be smoothly intercalated into interlayer I of $K_4Nb_6O_{17}$. [10, 14–17] The nanostructured hybrid $MV^{2+}-K_4Nb_6O_{17}$ was prepared by treating $K_4Nb_6O_{17} \cdot 3H_2O$ with an aqueous solution of excess methylviologen chloride and then allowing it to stand for 3 weeks at 70 °C. The resultant product was washed with deionized water until the MV^{2+} absorption could not be detected at 257 nm in the filtrate solution [22].

Fig. 2 The molecular structure (a) and structural model (b) of methylene blue cation

To make the $MB^+-K_4Nb_6O_{17}$ hybrid film, 150 µL aqueous suspension of $MV^{2+}-K_4Nb_6O_{17}$ composite was cast onto a quartz glass plate (20 × 40 mm²) to obtain an optically transparent film, and then the plate was placed in a 5 mM aqueous solution of methylene blue for 2 weeks. Rinsed the plate with deionized water carefully and dried to obtain a dark blue $MB^+-K_4Nb_6O_{17}$ nanocomposite thin film. All the procedures were prepared at room temperature. A thin film of $MB^+-K_4Nb_6O_{17}$ nanocomposite was prepared on the surface of a glass carbon electrode (GCE) with the same procedure for electrochemical investigation.

XRD patterns of the $MV^{2+}-K_4Nb_6O_{17}$ and $MB^+-K_4Nb_6O_{17}$ hybrid were collected with a M21X (MAC Co., Ltd.) diffractometer with monochromatic Cu K α radiation ($\lambda = 0.15406$ nm) with 2θ going from 1.5° to 40° in 1° steps. UV absorption spectra of the two hybrids were carried out using a UV–vis spectrometer (UV-2550). IR spectra were measured on a WGH-30/6 double-beam IR-spectrometer with the use of KBr pellets. Thermal gravimetric analysis (TGA) was carried out on a Shimadzu DTG-60 apparatus at a heating rate of 20 °C min⁻¹ from room temperature to 750 °C in air. Elemental analysis (EA) was performed using a Perkin Elmer 2400-CHN elemental analyzer.

The electrochemical experiments were carried out in a conventional three-electrode electrochemical cell at room temperature, with a platinum electrode as the counter electrode and a saturated calomel electrode (SCE) as the reference electrode. The $MB^+-K_4Nb_6O_{17}$ nanocomposite thin film modified GC electrode was used as the working electrode. The acting electrolyte was 0.1 mol L⁻¹ HCl solution. The solution was bubbled with N₂ for 20 min before examination in order to avoid the influence of oxygen. CV scans were carried out on a CHI600b electrochemical workstation at scan rate of 10–500 mV s⁻¹.

Results and discussion

Characterization of MB⁺–K₄Nb₆O₁₇ nanocomposite

 $K_4Nb_6O_{17} \cdot 3H_2O$ was identified by powder X-ray diffraction analysis (Fig. 3a). Sharp peaks indicated the high crystallinity of the $K_4Nb_6O_{17} \cdot 3H_2O$. The d_{020} value is





Fig. 3 The X-ray diffraction patterns of a $K_4Nb_6O_{17} \cdot 3H_2O$, b $MV^{2+}-K_4Nb_6O_{17}$ thin film, and c $MB^+-K_4Nb_6O_{17}$ thin film

regarded as basal spacing corresponding to the sum of the two adjacent interlayer spaces. The XRD pattern of hydrated potassium niobate $K_4Nb_6O_{17} \cdot 3H_2O$ (Fig. 3a) exhibited a (020) weak diffraction peak at 1.88 nm and an intense peak (040) at 0.94 nm, indicating that water molecules are intercalated only into the interlayer I [18]. The basal spacing of $MV^{2+}-K_4Nb_6O_{17}$ (Fig. 3b, 2.26 nm) was larger than that of $K_4Nb_6O_{17} \cdot 3H_2O$, indicating the substituent of MV^{2+} cations for water molecular in interlayer I. The interlayer I expansion of $MV^{2+}-K_4Nb_6O_{17}$ was calculated as 1.03 nm by subtracting the sum of thickness of the interlayer II and niobate layer of $K_4Nb_6O_{17} \cdot 3H_2O$ (1.23 nm) from the observed basal spacing (2.26 nm) [12].

As was shown in Fig. 3c, the 2θ angle of the (020) diffraction peak of MB⁺–K₄Nb₆O₁₇ was smaller than that of the mediate MV²⁺–K₄Nb₆O₁₇, showing that the intercalation of MB⁺ enlarged the interlayer I spacing of K₄Nb₆O₁₇. The interlayer I spacing of MB⁺–K₄Nb₆O₁₇ was calculated as 1.71 nm, which is larger than the spacing of MV²⁺–K₄Nb₆O₁₇ deposited from ESD precipitations [34]. Considering the rectangular dimensions of MB molecular ($1.70 \times 0.76 \times 0.325 \text{ nm}^3$) [36], we estimate that MB⁺ cations are placed in interlayer I in two ways, which is shown in Fig. 4. The MB⁺ cations may form a double layer parallel to the [Nb₆O₁₇]⁴⁻ layers, or a single layer standing vertically to the [Nb₆O₁₇]⁴⁻ layers.



Fig. 5 IR spectra of a $K_4 N b_6 O_{17}$ b MB, and c $MB^+{-}K_4 N b_6 O_{17}$ nanocomposite

The intercalation of MB into of K₄Nb₆O₁₇ was confirmed by IR spectra in Fig. 5. Characteristic infrared absorption peaks of MB [27, 29, 37] in Fig. 5b at 1604 cm⁻¹ and 1494 (stretching modes of the aromatic rings), 1400 cm^{-1} (C–N symmetric stretch), 1358 cm⁻¹ (-CH₃ symmetric deformation), and of $K_4Nb_6O_{17}$ [38, 39] at 905 cm⁻¹ and 539 cm^{-1} (Nb–O stretching vibration) in Fig. 4a appeared in the IR spectrum of MB⁺-K₄Nb₆O₁₇ (Fig. 5c). This confirms that MB has been intercalated into the interlayer spaces of niobate successfully without disturbing the structure of MB. However, there were some differences between the IR spectra of $MB^+-K_4Nb_6O_{17}$ and the spectra of MB and $K_4Nb_6O_{17}$. For MB, the adsorption band at 1604 cm⁻¹ shifted to 1608 cm^{-1} in the IR adsorption of the nanocomposites, for $K_4Nb_6O_{17}$, the adsorption band at 539 cm⁻¹ shifted to 529 cm^{-1} . It can be contributed to the bonding interaction between the Nb-O groups of the niobate and the nitrogen of the aromatic ring of MB [27, 38].

On the base of the elemental analysis data (C = 10.17%, H = 1.38%, N = 2.25%), the molecular formula of the MB⁺–K₄Nb₆O₁₇ nanocomposite is assigned as MB_{0.63}–K_{3.37}Nb₆O₁₇ · 2.55H₂O. The C/N mole ratio of the CHN analysis is calculated as 5.27, which is in good agreement with the observed value, 5.33, proves that the methylviologen ions in the niobate interlayer were replaced by MB⁺. Figure 6 gives the TG curve of the







Fig. 6 TGA curve of MB⁺-K₄Nb₆O₁₇ nanocomposite

 $MB^+-K_4Nb_6O_{17}$ nanocomposite. We explained the thermal behavior of the $MB^+-K_4Nb_6O_{17}$ nanocomposite with a two-step weight loss process. The first weight loss step from room temperature to 330 °C is caused by the release of intercalated water, which is dependent on the ambient conditions because of the sensitivity of $K_4Nb_6O_{17}$ to the atmospheric humidity [25, 26]. The second weight loss above 330 °C is due to the decomposition of the organic portion in the nanocomposite interlayer. The total weight loss (19%) is consistent with the sum of the water and MB^+ content determined by the elemental analysis.

The charge density of the $[Nb_6O_{17}]^{4-}$ layer is 0.126 nm² per negative charge [16], so the projected area of each MB⁺ ion in interlayer I can be calculated as $0.126 \times 2/0.63 = 0.4 \text{ nm}^2$. Judging from the dimensions of MB molecular, it can be concluded that the intercalation models in Fig. 4 (double layer parallel/monolayer perpendicular) are reasonable.



Fig. 7 UV spectrum of $MV^{2+}\text{-}K_4Nb_6O_{17}$ (a) and $MB^+\text{-}K_4Nb_6O_{17}$ (b) nanocomposites

UV-vis optical absorption of MB⁺-K₄Nb₆O₁₇ nanocomposite is shown in Fig. 7b, the UV spectrum of MV^{2+} -K₄Nb₆O₁₇ mediate (Fig. 7a) is also given for comparison. There is a broad absorption band in the visible light region in curve b, presenting a maximum absorbance at 586 nm and a shoulder at 680 nm, which confirms the presence of MB⁺ in interlayer I of K₄Nb₆O₁₇. It is known that MB⁺ has strong tendency to aggregate in aqueous solution, the typical maximum absorbance peaks locate at around 665 nm and 610 nm, ascribing to monomer and dimer, respectively [27, 40–44]. The shift of the MB monomer (665 nm) peak toward shorter wavelength clearly denotes that MB^+ cations in K₄Nb₆O₁₇ interlayer are aggregated. Considering the position of the maximum (586 nm) and the shape of the signal, we suggest that MB is mainly present as trimers and dimers, while the proportion of monomers is relatively low [36, 45]. This result is corresponding to the $MB^+-K_4Nb_6O_{17}$ nanocomposite obtained by ESD [34].

Electrochemical behavior of $MB^+-K_4Nb_6O_{17}$ nanocomposite thin film

The CV curve of MB in aqueous solution at 50mV s⁻¹ scan rate is shown in Fig. 8a. There are a couple of sensitive oxidation/reduction peaks with redox potentials at 0.205 V and 0.174 V, with the midpoint potential $[E_{\rm m} = (E_{\rm pa} + E_{\rm pc})/2]$ of 0.190 V. The peak separation $[\Delta E_{\rm p} = (E_{\rm pa} - E_{\rm pc})]$ is 31 mV, which is between the $\Delta E_{\rm p}$ value of individual electron transfer reaction (59 mV) and two-electron transfer reaction (28.5 mV), indicating a two-step individual electron transfer reactions of MB dimer [46, 47].

The cyclic voltammogram of $MB^+-K_4Nb_6O_{17}$ nanocomposite thin film on GCE is shown in Fig. 8b. The redox



Fig. 8 Cyclic voltammograms of a MB (0.05 mmol L^{-1}) and b MB⁺-K₄Nb₆O₁₇ thin film in 0.1 mol L^{-1} HCl solution, scan rate: 50 mV s⁻¹



Fig. 9 CVs of $MB^+-K_4Nb_6O_{17}$ thin film at different scan rates (10, 20, 30, 40, 50, 60, 70, 80, 100, 150, 200, 250, 300, 400, 500 mV s⁻¹ from inner to outer)

potentials at 0.202 V and 0.148 V with a midpoint potential of 0.175 V are similar with the peaks of MB in solution except the larger peak separation of 54 mV. There is a shift of E_{pc} to more negative values and a shift of E_{pa} to more positive values with increasing the scan rate (Fig. 9). We describe this redox peaks as a two-electron quasi-reversible redox process of MB dimer [31, 48–50]. The linear dependence of the cathodic peak current (I_c) on the square root of the scan rate (Fig. 10) displays a planar diffusion controlled reduction behavior of MB⁺ cations in niobate interlayer. The ΔE_p increased from 22 mV up to 143 mV when the scan rate varied from 10 to 500mV s⁻¹, indicating a slow electron diffusion process of the MB⁺ cations in the interlayer space at high scan rates. This is due to the semiconducting nature of the niobate matrix [31].

The ideal linear relationship in Fig. 10 indicates a fine mass transfer process, which is suitable for utilization as



Fig. 10 $I_c \sim v^{1/2}$ curve for MB⁺–K₄Nb₆O₁₇ layer nanocomposite thin film in 0.1 mol L⁻¹ HCl solution

electrode modifying. In order to test the electrochemical stability of the MB^+ – $K_4Nb_6O_{17}$ hybrid film, the modified GCE was tested again after exposure in air for 7 days. The cathodic and anodic currents of redox peaks I were still 90% of initial value at almost the same peak potential, confirming the good immobilization of MB in niobate interlayer spacings.

Conclusion

Laminar nanomaterial K₄Nb₆O₁₇ was synthesized through solid-phase synthesis and characterized by XRD. Methvlviologen was intercalated into interlayer I of K₄Nb₆O₁₇ by ion exchange reaction. Then methylene blue (MB) substituted the MV²⁺ cations through guest-guest ion exchange method. The two hybrid nanocomposites were characterized using IR, UV, XRD. UV pattern of MB⁺/ $K_4Nb_6O_{17}$ thin film indicated that the intercalated MB⁺ existed mainly as dimer and trimer, while the portion of monomer is fairly low. The cyclic voltammogram of the $MB^+-K_4Nb_6O_{17}$ nanocomposite film exhibited a pair of distinct reductive and oxidative peaks, representing a twoelectron redox process of MB dimer. The cathodic current exhibited a fine diffusion-controlled process, and the electrochemical stability of the hybrid film was also proved. We predict that novel MB⁺-K₄Nb₆O₁₇ nanocomposite has possibility to be used as electrode modifying material.

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